

# Fabrication of Metallic Nanodots in Large-Area Arrays by Mold-to-Mold Cross Imprinting (MTMCI)

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## ABSTRACT

We have developed a mold-to-mold cross imprint (MTMCI) process, which redefines an imprint mold with another imprint mold. By performing MTMCI on two identical imprint molds with silicon spacer nanowires in a perpendicular arrangement, we fabricated a large array of sub-30-nm silicon nanopillars. Large-area arrays of Pt dots are then produced using nanoimprint lithography with the silicon nanopillar mold.

Recently, great effort has been expended to develop facile and reliable techniques to fabricate periodic nanostructures because of their potential impacts on data storage,<sup>1–3</sup> photonic crystals,<sup>4,5</sup> biological sensors<sup>6–8</sup> and model catalysts.<sup>9</sup> Electron beam lithography (EBL)<sup>10</sup> and focused ion beam lithography (FIB)<sup>11</sup> are the most common techniques used to fabricate sub-50-nm patterns, but their low throughput limits their application, especially when large-area patterning is required. Thus, unconventional lithographic methods for producing periodic nanostructures with high densities, large areas, and small feature sizes are being pursued actively.<sup>12–16</sup> Nanoimprint lithography (NIL)<sup>17</sup> is a very promising alternative because it can replicate nanoscale features with reasonable cost and throughput. State-of-the-art NIL can even replicate the features of a mold with sub-10-nm resolution. However, the technology is still fundamentally dependent on a primary pattern generation technique, such as electron or ion-beam lithography, to generate the template.

Recently, a residual conformal film on the edge of a photolithographically defined pattern (a so-called “spacer”) has been used to generate nanoscale line features, with a line width that is controlled accurately by the deposited film thickness.<sup>18,19</sup> These nanoscale line patterns generated by this technique, termed spacer lithography, were originally de-

veloped to fabricate nanotransistors. This approach was adapted quickly to produce an imprint mold for metal nanowires.<sup>20</sup> The spacer technique enables the batch fabrication of sub-20-nm line features over wafer-scale areas using conventional optical lithography with high throughput. Note that spacer lithography is not throughput-limited in the way that high-resolution serial patterning methods such as EBL or FIB are because of its use of optical lithography, which is a parallel process. Multiple iterations of the spacer process (size-reduction lithography) enable the fabrication of features with a pitch far less than that of the initial photolithography.<sup>18,19</sup> However, these techniques are limited to fabricate closed-loop-shaped line patterns. In particular, discrete dot patterns cannot be produced because spacers are formed along the sidewalls of the original features.

In this letter, we report a complete parallel process for making an imprint mold with sub-30-nm periodic dot structures by combining spacer lithography and NIL. The silicon nanowires formed using spacer lithography are converted into silicon nanopillars by our newly developed mold-to-mold cross imprint (MTMCI) process. The resulting silicon nanopillars with dimensions of 15 × 30 × 100 nm (length × width × height) are then used as an imprinting mold, and Pt nanodot patterns have been replicated from the Si pillars to Pt dots on SiO<sub>2</sub>, which could serve as two-dimensional model catalysts.<sup>21,22</sup>

**Fabrication of a Single-Crystal Si Wire Mold by Spacer Lithography.** Silicon nanowire molds were fabricated at wafer scale by spacer lithography. The process is shown

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nanowires have the dimensions of  $15\text{ nm} \times 100\text{ nm} \times 12\text{ }\mu\text{m}$  (width  $\times$  height  $\times$  length).

**Fabrication of a Single-Crystalline Si Nanopillar Mold by Mold-to-Mold Cross Imprint (MTMCI).** To fabricate a large-area silicon nanopillar mold, we have developed the mold-to-mold cross imprint (MTMCI) technique, which converts the silicon spacer nanowires into silicon nanopillars. The concept is to further pattern a silicon spacer nanowire mold with another spacer nanowire mold by imprinting one mold placed perpendicular onto the other. The principle difference between MTMCI and conventional NIL is that the substrates for this imprinting is not flat and will eventually be processed further to become a mold for the second imprint.

The MTMCI fabrication process is shown in Figure 2a. First, the two silicon spacer nanowire molds with the dimensions of  $15\text{ nm} \times 100\text{ nm} \times 12\text{ }\mu\text{m}$  (width  $\times$  height  $\times$  length) are prepared. One is a bottom mold and the other is a top mold as named in Figure 2a. A 320-nm-thick film of PMMA is spin-cast on the bottom mold. PMMA (MW = 15 k) is dissolved in toluene at room temperature for 24 h. Before use, the solution is filtered by a  $0.2\text{-}\mu\text{m}$  PTFE filter and ultrasonicated for 3 h to remove any dissolved gas in the solution. The film thickness is controlled by both the PMMA concentration and spin rate. A typical 320-nm-thick film is obtained under conditions of 8 wt % and 1500 rpm. Immediately after a film is spin-cast, it is baked at  $150\text{ }^{\circ}\text{C}$  for 5 min. The PMMA thickness should be at least twice as large as the height of the spacer nanowire to cover the surface topography of the bottom mold and prevent mechanical damage of the wires during the imprint process. Thicker PMMA films can be used to improve the imprint uniformity, but at the cost of a more difficult breakthrough etch.

The top silicon mold is coated with tridecafluoro-1,1,2,2-tetrahydrooctyl-1-trichlorosilane (FTS) to form a fluorine-terminated self-assembled monolayer (SAM) on the surface. To form the SAM layer, the mold is first cleaned in piranha solution (a 7:3 mixture of 98%  $\text{H}_2\text{SO}_4$  and 30%  $\text{H}_2\text{O}_2$ ) at  $120\text{ }^{\circ}\text{C}$  to remove the organic contaminants and to form a thin layer of surface oxide and silanol groups on the mold surface. Then the clean mold is transferred into an oven filled with 2 mTorr FTS at  $90\text{ }^{\circ}\text{C}$  for 200 s. Functionalizing the surface lowers the surface energy in comparison with the PMMA substrate, thereby facilitating the separation of top mold from the PMMA after imprinting.

Next, the top mold and the bottom mold, coated with PMMA and FTS, respectively, are aligned so that the nanowire patterns in each mold are perpendicular to each other. Up to this step, all processes are performed in a class-100 clean room environment to avoid possible dust contamination, which may ruin the conformal contact between the molds. The wire patterns in the top mold are cross-imprinted into the PMMA layer on the bottom mold to form narrow trenches. The samples are imprinted under 4000 PSI at  $120\text{ }^{\circ}\text{C}$  for 300 s using a homemade press.<sup>23</sup> The imprint chamber with a diameter of 64 mm is heated resistively from room temperature to the imprinting temperature at a ramp rate of  $2\text{ }^{\circ}\text{C}/\text{min}$ . The cell is vacuum-pumped to remove the

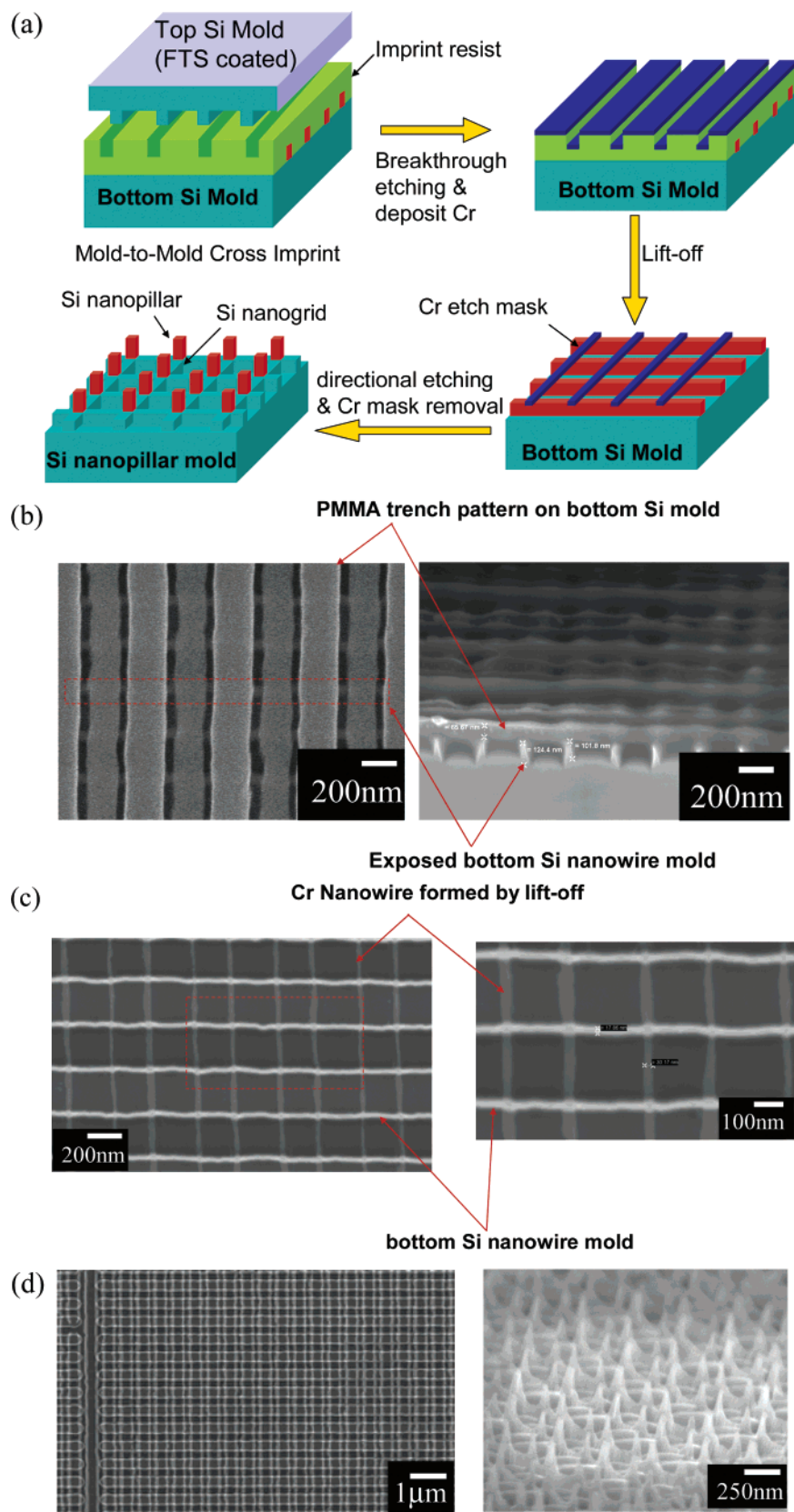
vapors that the sample releases during heating and pressing. After the imprint, the sample is air cooled to room temperature under pressure, and the substrate and mold are separated manually. Because the top mold is coated by FTS, the patterned PMMA is left on the bottom nanowire mold. In this way, PMMA nanotrenches are formed on top of and perpendicular to the silicon nanowires of the bottom mold.

After separating the two molds, the residual PMMA layer at the bottom of the imprinted trenches is subjected to a breakthrough etch to expose a portion of the underlying Si nanowires. There is an isotropic component to the PMMA etching, which leads to an increase of the trench width in the PMMA and thus affects the final feature sizes of the pillars. In addition, aspect-ratio-dependent etching effects cause the etch rate at the PMMA top surface to be higher than that at the bottom of the trench. To minimize the broadening during the breakthrough etching and protect the top surface of PMMA, a 10-nm-thick chromium layer is evaporated with an angle of  $45^{\circ}$  with respect to the surface normal. This angled evaporation enables us to coat selectively the top surface of the PMMA while leaving the bottom of the PMMA trenches unaffected. Breakthrough etching is performed with an  $\text{O}_2$  plasma at  $-100\text{ }^{\circ}\text{C}$ . The cryo-cooling minimizes the isotropic component of the etch and reduces trench broadening. The top-view and cross-section SEM images of imprinted PMMA trenches are shown in Figure 2b. The PMMA trenches are perpendicular to the bottom silicon nanowires so that only a small portion of them is exposed. This exposed region is where the silicon nanopillar will eventually be formed.

A layer of 10-nm-thick chromium is deposited by e-beam evaporation through the PMMA trenches onto the Si nanowires exposed by breakthrough etching. After lift-off of the PMMA in dichloromethane with ultrasonication, chromium nanowires anchored on the exposed part of the silicon nanowires are made. The chromium nanowires are perpendicular to the silicon spacer nanowires as shown in Figure 2c. The chromium nanowires are wider than the original silicon spacers on the top mold because of the trench broadening that occurs during the breakthrough etching and FTS coating.

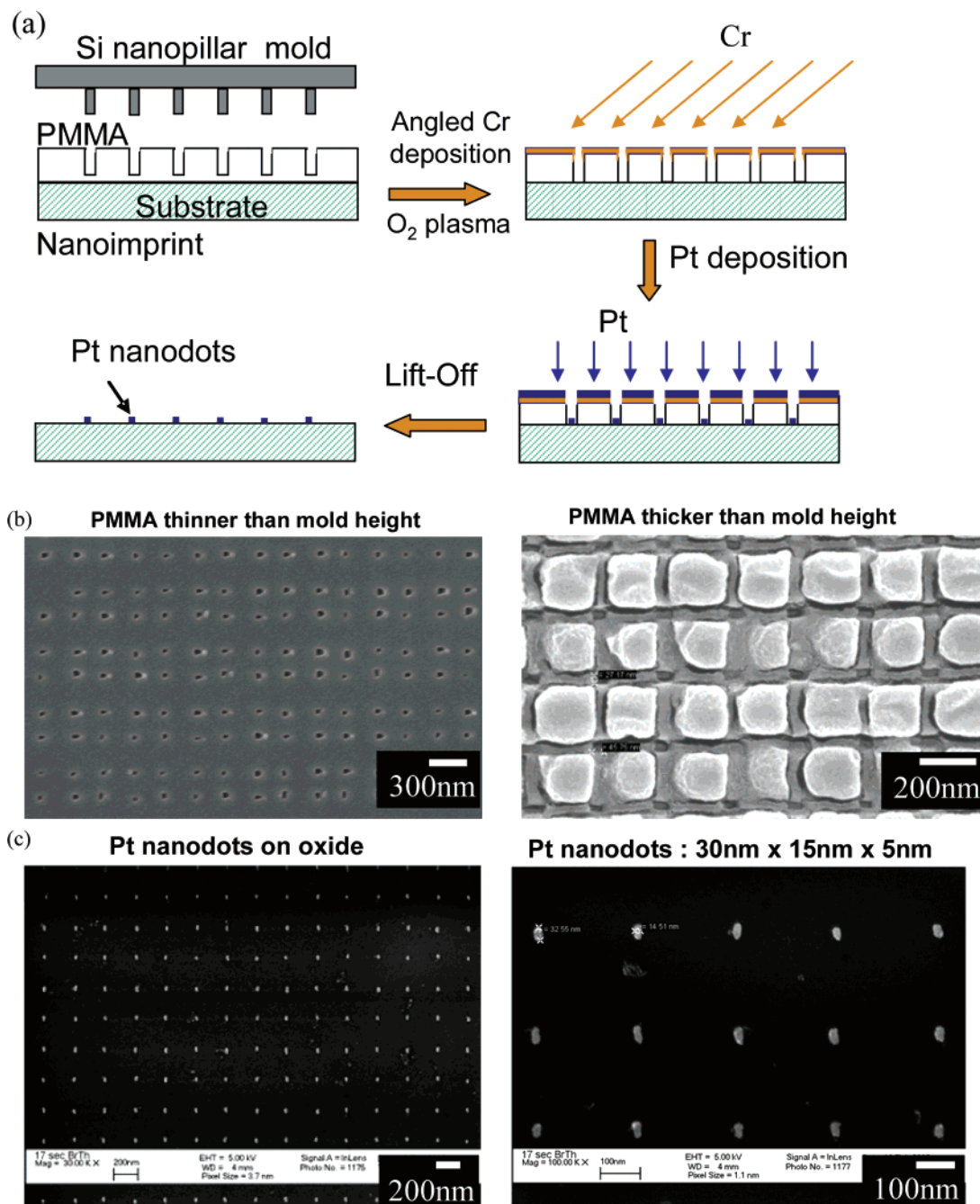
Finally, the silicon nanopillars are formed at the intersection of the silicon and chromium nanowires by etching the silicon using the chromium nanowire as an etch mask. The resulting silicon nanopillar mold is shown in Figure 2d. The final rectangular silicon nanopillars are 100-nm high with a top-view dimension of  $15 \pm 0.5\text{ nm}$  by  $30 \pm 2.3\text{ nm}$ . The pitch of the nanopillars is 250 nm, and areas of  $25\text{ cm}^2$  are patterned uniformly.

The MTMCI process described above produced a new mold with a dot pattern, a binary AND of the two original molds. In other words, the nanowire pattern in the original mold is converted to a nanopillar pattern. In addition, a new mold with a dashed line pattern, a binary XOR of the two original molds, can also be generated by using the imprinted PMMA trench as an etch mask for silicon instead of a lift-off mask (see Figure 3b). Although this technique was originally envisioned as a method for producing regular



**Figure 2.** Mold-to-mold cross imprint (MTMCI). (a) Schematic of the MTMCI process for converting nanowires into nanopillars. A silicon spacer nanowire mold is redefined with another spacer nanowire mold by imprinting one mold onto the other. (b) SEM top view (left) and cross section view (right) of the PMMA trenches on the bottom mold after imprinting. The PMMA trenches are perpendicular to the bottom silicon nanowires; therefore, only a small portion of bottom silicon nanowire is exposed. This exposed pattern will become silicon nanopillars eventually. (c) SEM images of the chromium nanowires anchored on the exposed part of the silicon nanowires. The chromium nanowires are perpendicular to the silicon spacer nanowires, thus making a grid pattern together with the underlying silicon nanowires. The chromium nanowires are wider than the silicon nanowires because of the trench widening during the PMMA breakthrough etching. (d) Top view (top) and angled view (bottom) of the silicon nanopillars. Uniform silicon nanopillars are formed at the intersections of the silicon and chromium nanowires shown in c. The nanopillars are on top of the nanogrids, making two vertical levels in the mold.





**Figure 3.** Imprinting of Pt nanodots. (a) Schematic of nanoimprint process for Pt dot formation. The silicon nanopillar mold from MTMCI is used as an imprint mold. (b) Imprinted hole pattern in PMMA when the PMMA resistor layer is thinner (left) or thicker (right) than the height of the silicon nanopillars. The latter has been used for Pt dot generation practically to prevent mechanical breakage of the mold. (c) Sub-30-nm Pt dots on the oxide surface after lift-off. Uniform Pt dot arrays are produced over a large area of 25 cm<sup>2</sup>.

arrays of nanoscale features, it is clear that the features produced by the initial lithographic step are not required to be regular. In addition, substantial variations in dot size are possible by appropriately choosing the thickness of the spacer. Pitches can be reduced further by means of size-reduction lithography.<sup>18,19</sup> Variation in feature size and spacing can be introduced at this level, again through appropriate choice of spacer and sacrificial layer thicknesses.

**Imprinting of Pt Nanodots.** Once the silicon nanopillar mold is fabricated by MTMCI, one can pattern various metal nanodots over a large area easily using the silicon nanopillars as a NIL mold. As a demonstration, Pt dots are patterned on

a flat oxide substrate. The silicon nanopillar mold is imprinted into a PMMA layer on silicon dioxide. Residual PMMA is removed by a breakthrough etch after deposition of a chromium hard mask by angled evaporation. Pt dots with dimensions of 15 × 30 × 5 nm (width × length × height) are patterned using e-beam evaporation and lift-off. The imprint process shown in Figure 3a is similar to that demonstrated previously for nanowire formation,<sup>20</sup> except that the breakthrough etching step is more challenging for dot patterning. The mold has two vertical steps because the 100-nm-tall silicon nanopillar is formed on top of the 100-nm-tall grid pattern. Therefore, two different PMMA patterns

can be made as shown in Figure 3b. When the thickness of PMMA for imprinting is less than the height of the silicon nanopillar, only the pillar is imprinted as a hole (Figure 3b left). If the PMMA is thicker than the height of the pillars, then nanoholes in PMMA will show up at the corners of the grids (Figure 3b, right). The breakthrough etching for the former is easier than the latter because of its lower aspect ratio and thinner residual layer. Breakthrough etching for the latter is more difficult because overetching will result in a grid pattern instead of a dot pattern (one can overetch intentionally to produce the grid pattern, that is, a binary OR pattern of the two original molds). However, we used a 250-nm-thick PMMA layer in order to prevent mechanical breakage of the mold. The PMMA is 50-nm thicker than the maximum height of the features on the mold, so direct physical contact between the substrate and the mold is avoided and results in better uniformity. Imprint in this manner leads to two vertical levels of the PMMA layer because of both the silicon nanopillars and grids. Then, timed breakthrough etching is performed to expose the pillar patterns only. Large arrays of well ordered sub-30-nm Pt dots are produced, as shown in Figure 3c. These Pt nanodots will be used as model heterogeneous catalysts.

**Summary.** We have demonstrated mold-to-mold cross imprint (MTMCI), a parallel process to fabricate sub-30-nm metal dots with high throughput by combining optical, spacer, and nanoimprint lithographies. By performing MTMCI on two identical imprint molds with silicon spacer nanowires in a perpendicular arrangement, we have fabricated large arrays of sub-30-nm silicon nanopillars. We note that substantial variations in dot size and pitch are possible by appropriately choosing the thickness of the spacer and initial line width of sacrificial layers. Large-area arrays of Pt dots are then imprinted using nanoimprint lithography with the silicon nanopillar mold. The metal nanodot arrays are well ordered because of the lithographic definition of the pitches, and their dimensions are controlled precisely to sub-30-nm by combination of the spacer lithography and nanoimprint. This technique is capable of producing high areal density nanodots of any metal on any surface.

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## References

- (1) Cheng, J. Y.; Ross, C. A.; Chan, V. Z. H.; Thomas, E. L.; Lammertink, R. G. H.; Vancso, G. J. *Adv. Mater.* **2001**, *13*, 1174.
- (2) Hehn, M.; Ounadjela, K.; Bucher, J. P.; Rousseaux, F.; Decanini, D.; Bartenlian, B.; Chappert, C. *Science* **1996**, *272*, 1782.
- (3) Wu, W.; Cui, B.; Sun, X. Y.; Zhang, W.; Zhunag, L.; Chou, S. Y. *J. Vac. Sci. Technol., B* **1998**, *16*, 3825.
- (4) Poborechii, V. V.; Tada, T.; Kanayama, T. *Appl. Phys. Lett.* **1999**, *75*, 3276.
- (5) Wanke, M. C.; Lehmann, O.; Muller, K.; Wen, Q. Z.; Stuke, M. *Science* **1997**, *275*, 1284.
- (6) Haes, A. J.; Van Duyne, R. P. *J. Am. Chem. Soc.* **2002**, *124*, 10596.
- (7) Lee, K. B.; Park, S. J.; Mirkin, C. A.; Smith, J. C.; Mrksich, M. *Science* **2002**, *295*, 1702.
- (8) Cui, Y.; Wei, Q. Q.; Park, H. K.; Lieber, C. M. *Science* **2001**, *293*, 1289.
- (9) Grunes, J.; Zhu, J.; Somorjai, G. A. *Chem. Comm.* **2003**, *19*, 2510.
- (10) Saitou, N. *Int. J. Jpn. Soc. Precis. Eng.* **1996**, *30*, 107–111.
- (11) Matsui, S.; Ochiai, Y. *Nanotechnology* **1996**, *7*, 247–258.
- (12) Fay, B. *Miroelectron. Eng.* **2002**, *11*, 61.
- (13) Gates, B. D.; Xu, Q.; Love, J. C.; Wolfe, D. B.; Whitesides, G. M. *Annu. Rev. Mater. Res.* **2004**, *34*, 339.
- (14) Younan, X.; Whitesides, G. M. *Angew. Chem., Int. Ed.* **1998**, *37*, 550.
- (15) Hamley, I. W. *Angew. Chem., Int. Ed.* **2003**, *42*, 1692–1712.
- (16) Suh, K. Y.; Choi, S.-J.; Baek, S. J.; Kim, T. W.; Langer, R. *Adv. Mater.* **2005**, *17*, 560.
- (17) Guo, L. J. *J. Phys. D: Appl. Phys.* **2004**, *37*, R123–R141.
- (18) Choi, Y. K.; Lee, J. S.; Zhu, J.; Somorjai, G. A.; Lee, L. P.; Bokor, J. *J. Vac. Sci. Technol., B* **2003**, *21*, 2951.
- (19) Choi, Y. K.; Zhu, J.; Grunes, J.; Bokor, J.; Somorjai, G. A. *J. Phys. Chem. B* **2003**, *107*, 3340.
- (20) Yan, X. M.; Kwon, S.; Contreras, A. M.; Bokor, J.; Somorjai, G. A. *Nano Lett.* **2005**, *5*, 745.
- (21) Contreras, A. M.; Grunes, J.; Yan, X. M.; Liddle, A.; Somorjai, G. A. *Catal. Lett.* **2005**, *100*, 115.
- (22) Yan, X. M.; Kwon, S.; Contreras, A. M.; Bokor, J.; Somorjai, G. A. *Catal Lett.*, in press.
- (23) Zhu, J. Fabrication, Characterization and Reaction Studies of Nanofabricated Platinum Model Catalysts. Ph.D. Dissertation, University of California at Berkeley, 2003.

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